

Simulation study on Stone wales defect in Si-Nanowire based Graphene-lithium ion batteries

Ankit Mishra¹, Ashutosh Choudhary², Sameera Khan³, Gagan Jain⁴

¹ Assistant Professor, Amity School of Engineering & Technology, Amity University Chhattisgarh

² Assistant Professor, Amity School of Engineering & Technology, Amity University Chhattisgarh

³ Assistant Professor, Amity School of Engineering & Technology, Amity University Chhattisgarh

⁴ Assistant Professor, Amity School of Engineering & Technology, Amity University Chhattisgarh

Abstract: In this paper we study of atomic scale properties of Si-Nanowire based graphene lithium ion batteries with stone-wales defect which is simulated by using Atomistix tool kit (ATK). The differences in the transmission spectrum, Fermi level and defects are discussed in detail. ATK is a research tool which provides proficient calculations of materials electron transport properties and atomic structures. The simulation results were obtained by using Extended Huckel approximation.

Index Terms— ATK, EXTENDED HUCKEL, Graphene,

I. INTRODUCTION

Nowadays, an increase of technology results in the increase of energy demands. High power batteries which can sustain the charge cycle for the long duration of time can be most prominently used. From last 5 years, Li-ions batteries are the most significantly used batteries in all electronics and electrical application [1]. These lithium ion batteries have many unique significant properties which can replace the conventional batteries. Advantages of these batteries are high energy density, no pollution, low self-discharging rate and long useful life span. Using of graphene as the anode material, increase the electronic property of battery which can further increase the efficiency of the battery. This honeycomb-like structure is one atom thick layer which has multiple properties in electronics, mechanical, chemical and many areas. It is used in solar cell, capacitors, biomedical, semiconductors and also used for high mechanical strengthen of materials [2]. This novel graphene material is chemically stable and due to the high electrical conductivity, this material is a most promising material which can be used as the alternative in the battery. In this paper we have, find the atomic property of graphene and lithium atom and also observe the transmission spectrum, conductance and Fermi energy of this material. Apart from this we also observe the atomic property when the material consists of some defect. Nanowires (NW) have shown great potential in low-cost high-performance solar cells[3-6]. Silicon (Si) NW solar cells are particularly promising due to the low cost, abundance in nature, nontoxicity, and long-term stability and have been widely reported in recent years [7–10]. However, due to its indirect bandgap, Si has a poor absorption of light at small thicknesses, particularly in the 600–1100-nm spectral range [11]. As the NW commonly has a short length of several microns, Si NW array solar cells typically have a low conversion Efficiency [7–9]. An efficient way of increasing the efficiency of Si NW solar cell is to enhance its absorption in the 600–1100-nm range via introducing new structures or materials. Recently, a family of methylammonium lead halide perovskites $\text{CH}_3\text{NH}_3\text{PbX}_3$ ($X = \text{I}, \text{Br}, \text{Cl}$) has attracted great attention for its breakthrough in low-cost high-efficiency solar cells [12–14]. The excellent properties including a direct bandgap, large absorption coefficient, long exciton diffusion length (100~1000 nm) and lifetime, and low cost processes make perovskite an ideal light absorber [12, 15, 16]. The perovskite materials typically have high and flat absorption efficiency in a wide spectrum.

II. DEVICE SIMULATION

Here we used Atomistix Tool Kit (ATK) for simulation and calculating the electrical transport property for graphene and lithium ion-atom. In ATK-DFT, semi-empirical tight bindings and classical function are used to simulate the device structure. In this, for simulating the nano scale device structure we use non – equilibrium greens function[17-19]. Here we used extended Hückel method which is the semi-empirical quantum chemistry method developed by the Roald Hoffmann. This method is used for determining the structural geometry of organic molecule

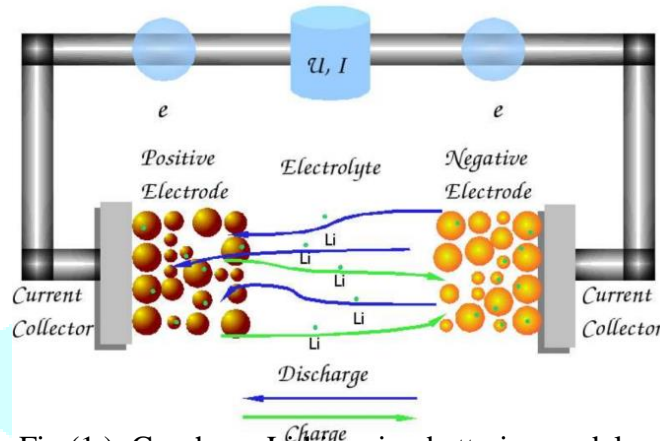


Fig (1) : Graphene-Lithium ion batteries model

and for molecular orbital's. It can calculate the relative energy of different geometrical structure, calculation of electronic interactions [3].

For solving the electronic structure we use basis of Hamiltonian local atomic orbital (LCAO)

$$\Phi_{nlm}(r) = R_{nl}(r)Y_{lm}(r) \quad (1)$$

Where Y_{lm} is a spherical harmonic, R_{nl} is a radial function.

$$S_{ij} = \int_v \Phi_i(r - R_i)\Phi_j(r - R_j)dr \quad (2)$$

Where R is the position of orbital center, i is composite index of nlm. For

calculating the basic function of integral form,

$$R_{nl}(r) = \frac{r^{n-1}}{\sqrt{(2n)!}} [C_1(2n_1)^{n+\frac{1}{2}}e^{-n_1r} + C_2(2n_2)^{n+\frac{1}{2}}e^{-n_2r}] \quad (3)$$

Where n_1, n_2, C_1 and C_2 are the adjustable parameters. These are used to defined angular shell of valence bond orbital's for each element [4].

Hamiltonian overlap matrix defines,

$$H_{ij} = \frac{1}{4}(\beta_i + \beta_j)(E_i + E_j)S_{ij} \quad (4)$$

Where E_i is the onsite orbital energy and β_i is a Hückel fitting parameter

DFT Calculation is used to investigate the electronic property of atomic structure.nical modelling method.The DFT calculate the Interacting of electrons at different state. The red balls showed the interaction of electron[21]. In DFT the density matrix is calculated by non-equilibrium Green's functions (NEGF)

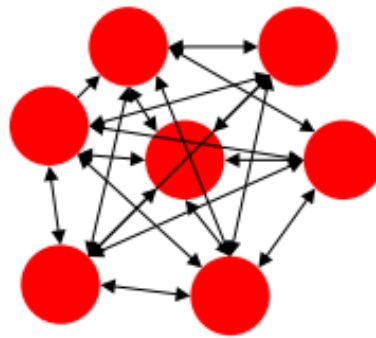
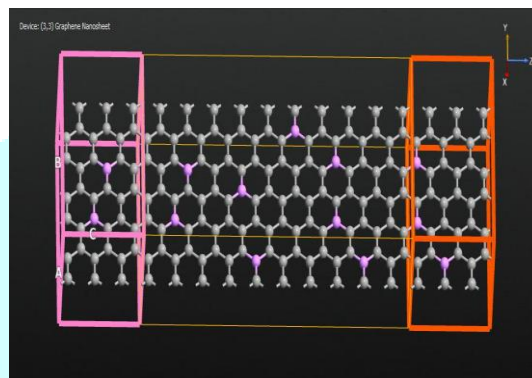


Fig (2) : Interaction of electron



III. FIGURES

Fig (3) :Si-Nanowire with no defect

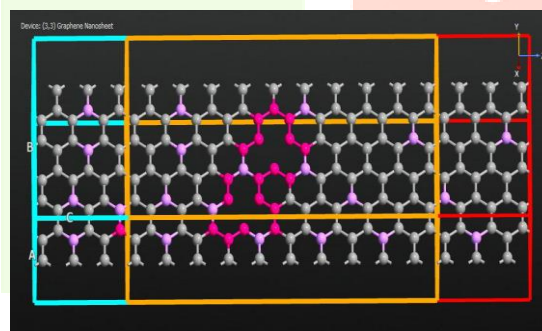


Fig (4) :Si-Nanowire with Stone-Wales defect

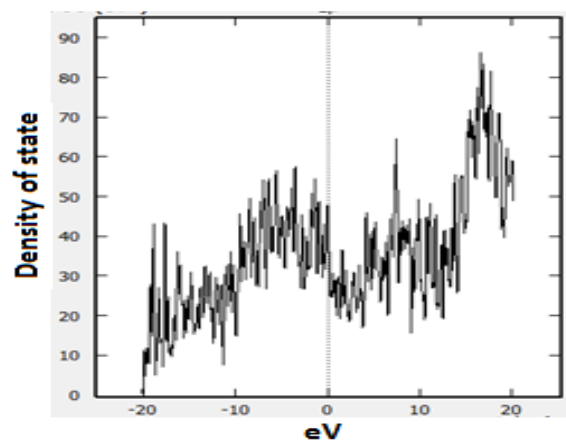


Fig (5): DOS VS eV with defect

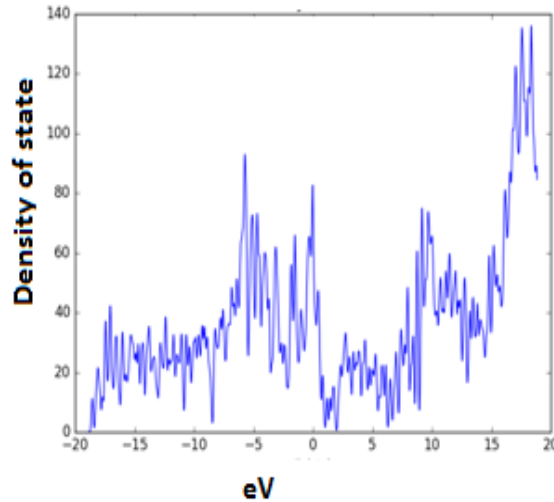


Fig (6): DOS VS eV with no defect

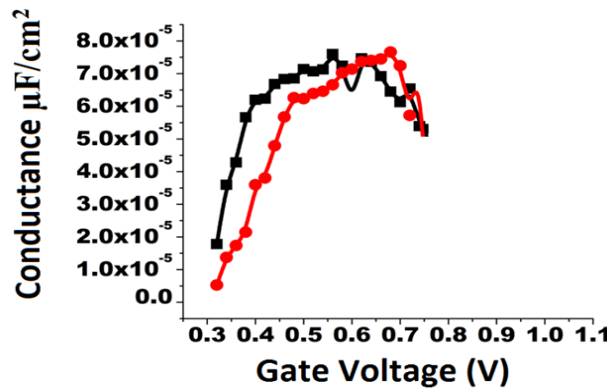


Fig (7): Conductance VS Gate Voltage

IV. RESULTS AND DISCUSSION

By using the Hückel semi empirical method, we observe that when the graphene and lithium ion atom is mix without any defect in there atomic structure the Fermi level for Left electrode = -3.190716×10^0 eV and for Right electrode Fermi level = 3.190716×10^0 eV. Whereas in the presence of a defect, the Fermi level is increased to the 4.46 eV. The density of state in fig(7) show the number of electron occupies in per state. In no defect area the electron occupies maximum number of state whereas in defect area the electron occupancy is less.

In no defect area the transmission of electron is high and covers the whole part as it can be seen in fig (24). Where as in defect area the transmission is high in starting area where as it get decreased slowly [25]. In fig (5), we will get higher spikes region, these is due to the in proper flow of electron where as in fig (6) the spikes are less as it doesn't have any defect in

atomic level. Fig 9 shows the electron density where in first part shows the higher number of electron whereas in second part shows least dense area due to the distortion in the atomic structure[28-30].

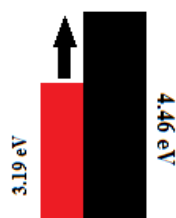


Fig (8): Increase in Fermi Level

(V) CONCLUSION

The extended Hückel method is the suitable for solving atomic scale structure and obtaining the electronic property of it. Here we observe that if there is no defect present in graphene then conductivity will be high whereas if there is any atomic defect present the there will change in the conductivity property. In fig (10), It is shown by using DFT and Extended Huckel calculation[31], the Fermi level of graphene – lithium ions increase upto 4.46 eV due to the structural defect.

VI. REFERENCE

1. F. Zahid, M. Paulsson, E. Polizzi, A. W. Ghosh, L. Siddiqui, and S. Datta, *J. of Chem. Phys.*, 123, 064707 (2005).
2. H. H. B. Sørensen, P. C. Hansen, D. E. Petersen, S. Skelboe, and K. Stokbro, *Phys. Rev. B*, 79, 205332 (2009).
3. *The transport calculations where formed with Atomistix ToolKit, version 2016. The manual is available online at <http://www.quantumwise.com/documents/manuals>.*
4. Kurt Stokbro, *Semi-Empirical Model for Nano-Scale Device Simulations*, 1004.2812v1, 2010
5. Qian Cheng, Zeming Song, Teng Ma, Bethany B. Smith, Rui Tang, Hongyu Yu, Hanqing Jiang, and Candace K. Chan, *Folding paper-based lithium-ion batteries for higher areal energy densities*
6. A. Kojima, K. Teshima, Y. Shirai, T. Miyasaka "Organometal halide perovskites as visible-light sensitizers for photovoltaic cells." *J. Am. Chem. Soc.* 2009, 131, 6050
7. Yixin Zhao and Kai Zhu "Charge Transport and Recombination in Perovskite (CH₃NH₃) PbI₃ Sensitized TiO₂ Solar Cells." *J. Phys. Chem. Lett.* 2013, 4, 2880–2884
8. Edoardo Mosconi, Anna Amati, Mohammad Khaja Nazeeruddin, Michael Grätzel, and Filippo De Angelis "First Principles Modeling of Mixed Halide Organometal Perovskites for Photovoltaic Applications" *J. Phys. Chem.* 2013, 117 (27), pp 13902–13913
9. ChogBarugkin, Jinjin Cong, The Duong, Shakir Rahman, Hieu T. Nguyen, Daniel Macdonald, Thomas P. White, and Kylie R. Catchpole "Ultralow Absorption Coefficient and Temperature Dependence of Radiative Recombination of CH₃NH₃PbI₃ Perovskite from Photoluminescence" 2015 *J. Phys. Chem. Lett.*, 6, 767–772
10. Robertson J and Clark S J 2011 Limits to doping in oxides *Phys. Rev. B* 83 075205
11. Malinkiewicz O, Yella A, Lee Y H, Espallargas G M, Graetzel M, Nazeeruddin M K and Bolink H J 2013 Perovskite solar cells employing organic charge-transport layers *Nat. Photonics* 8 128–32
12. Mrowec S and Grzesik Z 2004 Oxidation of nickel and transport properties of nickel oxide *J. Phys. Chem. Solids* 65 1651–7
13. Bi C, Shao Y, Yuan Y, Xiao Z, Wang C, Gao Y and Huang J understanding the formation and evolution of interdiffusion grown organ lead halide perovskite thin films by thermal annealing *J. Mater. Chem. A* 2 18508–14
14. Newman R and Chrenko R M 1959 Optical properties of nickel oxide *Phys. Rev.* 114 1507–13
15. Wehrenfennig C, Liu M, Snaith H J, Johnston M B and Herz L M 2014 Charge-carrier dynamics in vapour-deposited films of the organolead halide perovskite CH₃NH₃PbI₃-xCl_x *Energy Environ. Sci.* 7 2269–75
16. Wehrenfennig C, Eperon G E, Johnston M B, Snaith H J and Herz L M 2014 High charge carrier mobilities and lifetimes in organolead trihalide perovskites *Adv. Mater.* 26 1584–9
17. Hu L et al 2014 Sequential deposition of CH₃NH₃PbI₃ on planar NiO film for efficient planar perovskite solar cells *ACS Photonics* 1 547–53
18. Eperon G E, Burlakov V M, Docampo P, Goriely A and Snaith H J 2014 Morphological control for high performance, solution-processed planar heterojunction perovskite solar cells *Funct. Mater.* 24 151–
19. Saha S K, Guchhait A and Pal A J 2012 Cu₂ZnSnS₄ (CZTS) nanoparticle based nontoxic and earth-abundant hybrid pn junction solar cells *phys. Chem. Chem. Phys.* 14 8090–6
20. Blom P W M, De J M J M and Munster M G V 1997 Electric field and temperature dependence of the hole mobility in poly(p-phenylenevinylene) *Phys. Rev. B* 55 656–9
21. Deurmeier J, Gassmann J, Brötzel J, Klein A and Bro J 2011 Reactive magnetron sputtering of Cu₂O: dependence on oxygen pressure and interface formation with indium tin oxide *J. Appl. Phys.* 109 113704
22. Matsuzaki K, Nomura K, Yanagi H, Kamiya T, Hirano M and Hosono H 2008 Epitaxial growth of high mobility Cu₂O thin films and application to p-channel thin film transistor epitaxial growth of high mobility Cu₂O thin films and application to p-channel thin film transistor *appl. Phys. Lett.* 93 202107
23. Shi J et al 2014 Hole-conductor-free perovskite organic lead iodide heterojunction thin-film solar cells: high efficiency and junction property *appl. Phys. Lett.* 104 063901
24. Meyer B K et al 2012 Binary copper oxide semiconductors: from materials towards devices *Phys. Status Solidi* 249 1487–509
25. Conwell E and Weisskopf V F 1950 Theory of impurity scattering in semiconductors *Phys. Rev.* 77 388–90
26. Halme J, Vahermaa P, Miettunen K and Lund P 2010 Device physics of dye solar cells *Adv. Mater.* 22 210–34
27. Wang K, Shen P, Li M, Chen S, Lin M, Chen P and Guo T 2014 Low-temperature sputtered nickel oxide compact thin film as effective electron blocking layer for mesoscopic NiO/CH₃NH₃PbI₃ perovskite heterojunction solar cells *ACS Appl. Mater. Interfaces* 6 11851–8
28. Xing G, Mathews N, Lim S S, Yantara N, Liu X, Sabba D, Grätzel M, Mhaisalkar S and Sum T C 2014 Low temperature solution-processed wavelength-tunable perovskite for lasing *Nat. Mater.* 13 476–80
29. Jung J W, Williams S T and Jen A K-Y 2014 Low-temperature processed high-performance flexible perovskite solar cells via rationally optimized solvent washing treatment *R. Soc. Chem. Adv.* 4 62971–7
30. Abate A, Saliba M, Hollman D J, Stranks S D, Wojciechowski K, Avolio R, Grancini G, Petrozza A and Snaith H J 2014 Supramolecular halogen bond passivation of organic-inorganic halide perovskite solar cells *Nano Lett.* 14 3247–54
31. Noel N K, Abate A, Stranks S D, Parrott E, Burlakov V, Goriely A and Snaith H J 2014 Enhanced photoluminescence and solar cell performance via Lewis base passivation of organic-inorganic lead halide perovskites *ACS Nano* 8 9815–217 *Semicond. Sci. Technol.* 30 (2015) 054004 Y Wang et al